Understanding Charge Transport in Endohedral Fullerene Single-Crystal Field-Effect Transistors

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Abstract: The encapsulation of nitrogen within C_{60} forms non-metallic endohedral fullerene N@C₆₀. Previous calculations show that the encapsulated nitrogen may favor more efficient charge injection and transport under external electric fields when compared to C_{60} , suggesting that N@C₆₀ may be a promising candidate for application in organic electronic devices. However, owing to difficulties in both synthesis and purification, the potential application of N@C₆₀ under external electric field has not been previously studied experimentally and its intrinsic charge transport mechanism remains unknown, which hinders more widely applications of endohedral fullerene in organic electronic devices. Here, we demonstrate the field-effect study and photo-detective applications of solution-grown N@C₆₀ single crystals. Organic fieldeffect transistors (OFETs) based on them exhibit electron mobilities up to 2.23 cm² V⁻ ¹ s⁻¹. Furthermore, the electrical properties show a favorable band-like charge transport mechanism from 180K to 300K, and photodetectors based on them yield a highly sensitive photo-conductive property under the near-infrared (NIR) illumination with a responsivity of 177.3 A W⁻¹. This study, which outlined the intrinsic charge transport properties of N@C₆₀, should not only enable significant advancements for the highmobility n-type OFETs and highly sensitive photosensing applications, but also provide a reference for studying fundamental physics of endohedral fullerene.

Introduction

Organic field-effect transistors (OFETs) have attracted continuous research attention for flexible and light-weight electronic devices, such as circuits, ^{1–4} displays,⁵ sensors,^{6–10} and photodetectors.^{11–14} Among them, photodetectors, which translate optical signal into electrical signal, playing a vital role in high-resolution imaging, lightwave communications, and optical interconnection systems.^{15–20} Organic single crystals, being free of grain boundaries and molecular disorders, are regards as promising charge transport media for organic electronic devices.^{21,22} Fullerene C₆₀, is a superstar organic semiconductor for organic single crystal electronics, as it has achieved the highest electron mobility among organic semiconductors²³ and the photodetectors based on it exhibit highly sensitive photo-responsivity.^{14,24}

In recent years, endohedral fullerenes have attracted broad attentions due to their unique forms and new properties that are unexpected for empty fullerenes.²⁵ Endohedral fullerenes can be classified into endohedral metallofullerenes and non-metallo fullerenes according to different atoms enclosed.²⁶ Endohedral metallofullerene

OFETs have been fabricated with $Dy@C_{82}$, $La_2@C_{80}$ and so forth.^{27,28} However, owing to difficulties in the formation of crystals of endohedral metallofullerenes, they have only be applied in OFETs as amorphous films and the electron mobility (μ) values of these OFETs present only around 10⁻⁴ cm² V⁻¹ s⁻¹.^{27,28} The encapsulation of nitrogen within C₆₀ forms non-metallic endohedral fullerene N@C₆₀. It is stable because the location of the atomic nitrogen is in the center of the C₆₀ cage and it is well protected from the ambient environment.^{29,30} This property enables N@C₆₀ to form single crystals as easily as C₆₀. Previous calculations show that the encapsulated nitrogen decreases the lowest unoccupied molecular orbital (LUMO) energy level of C₆₀, which may favor more efficient charge injection and transport under external electric fields when compared to C_{60} .^{31,32} This ease of crystal formation, when coupled to efficient electron transport properties suggests that $N@C_{60}$ may be a promising candidate for FET applications. Unfortunately, owing to difficulties in both synthesis and purification, the potential application of $N(a)C_{60}$ under external electric field has not been previously studied experimentally.

Herein, we report the first demonstration of N@C₆₀ single crystal OFETs and highly sensitive NIR photodetectors. These are based on solution-grown onedimensional single crystals of N@C₆₀. The morphologies and crystal structures are studied by atomic force microscope (AFM), scanning electron microscope (SEM) and selected area electron diffraction (SAED) along with high-resolution transmission electron microscope (TEM). OFETs based on N@C₆₀ single crystals present μ as high as 2.23 cm² V⁻¹ s⁻¹. A temperature dependence study shows a band-like electron mobility mechanism, which is the signature of high-quality crystals and excellent electrical conduction. Photodetectors based on N@C₆₀ single crystals present highly sensitive photoconductivity properties with a high responsivity of 177.3 A W⁻¹ under NIR illumination, which is among the highest for the reported organic photodetectors working under similar conditions. The high mobility and highly sensitive photo-responsivity of the N@C₆₀ single crystals will accelerate the development of the high-performance OFETs and next-generation photodetection devices.

Results and Discussions

 $N@C_{60}$ was produced by nitrogen-ion implantation of C_{60} films and was purified by a two-step high-performance liquid chromatography (HPLC) process as reported in detail by us previously.³³ The droplet-pinned crystallization (DPC) is reported to be a simple and efficient method to align large-scale C_{60} single crystals.²³ The DPC method allows the crystals to grow along a substrate surface, with close contact between the single crystals and the substrates. Thus, it favors the formation of OFET charge transporting channels.^{34–36} Using this method, self-aligned one-dimensional $N@C_{60}$ crystals were successfully prepared (**Figure 1a**).

After the preparation of single crystals, the crystal morphologies and crystal structures were studied. To reduce dielectric surface traps, these were grown on octadecylsilanes (OTS) modified SiO₂/Si substrates, rather than bare SiO₂/Si substrates.^{37–40} The average dimensions were obtained based on 50 single crystals. Optical microscope (OM) shows that the obtained N@C₆₀ needle crystals had an average length of 294 µm with a standard deviation of 54 µm, *i.e.*, $(294 \pm 54 \mu m)$, which

is compared to that of reported C₆₀ needle crystals.²³ The SEM images, an example of which is shown in **Figure 1b**, indicates that the average width of $N(a)C_{60}$ crystal is (465) \pm 64) nm while AFM image shows a faceted shape and very smooth surfaces with average thickness of (285 ± 47) nm for N@C₆₀ needle crystal (Figure 1c), indicative of crystalline structures. To gain more insight into the crystal structures of the N@ C_{60} , the samples were characterized by transmission electron microscopy (TEM) and the selected area electron diffraction (SAED), as shown in Figure 1d. The SAED diffraction spot pattern is very similar to the reported crystal structure of solutiongrrown C_{60} ·*m*-xylene single crystals in the literature.²³ The SAED patterns from different areas of one single crystal are all identical, indicating the single-crystalline nature of as-prepared crystals. To further confirm the crystal structure, we performed an X-ray diffraction (XRD) study. The room-temperature XRD pattern of the asprepared single crystals is shown in Figure S1. The structure is indexed to be a hexagonal system and cell dimensions are a = 23.11 Å, b = 23.11 Å and c = 10.10 Å. This is similar to the reported C₆₀ nanorods achieving by slow solvent evaporation of an *m*-xylene solution of C₆₀ (a = 23.76 Å, b = 23.76 Å and c = 10.08 Å).⁴¹ Further information about the crystallinity was obtained from a high-resolution transmission electron microscopy (HRTEM) study (Figure S2.). Not only does this image confirm that the N@C₆₀ needles are single crystals, it also shows that the measured 1.16 nm spacing between neighbouring lattice planes equals to the distance between two (110) crystal planes obtained by XRD. Therefore, (001) is the preferential growth direction

of as-prepared N@C₆₀ crystals. These morphology studies indicate that high quality N@C₆₀ single crystals are successfully obtained.

OFETs were fabricated based on as-grown single crystal needles in a top-contact bottom-gate device configuration (Figure 2a). The devices were tested under vacuum, and the saturation region electron mobility was obtained. The channel length and width were obtained based on SEM images as shown in Figure 2b. The transfer and output characteristics of typical n-channel OFETs were obtained (Figure 2c and 2d), exhibiting good gate modulation. The mobility was gate bias-dependent (Figure S3), and the mobility was extracted following the method reported in the literature.⁴² At a $N@C_{60}$ concentration of 0.5 mg mL⁻¹, the highest electron mobility (μ) of 2.23 cm² V⁻ ¹ s⁻¹, on/off current ratio >10⁴, and threshold voltage ($V_{\rm T}$) of 19.52 V were achieved (Figure 2c). As the measurement of contact channel width can strongly influence the measurement of charge carrier mobility, we therefore measured the channel length of 100 devices and summarized the relevant mobilities in Figure S4. Based on these 100 devices, the average μ is (1.84 \pm 0.26) cm² V⁻¹ s⁻¹. To the best of our knowledge, the mobility of our N@C₆₀-based OFETs is the highest for OFETs based on endohedral fullerenes.27,28

OFETs based on N@C₆₀ single crystals grown from solutions of varied N@C₆₀ concentrations were then examined. Different concentrations of N@C₆₀ in *m*-xylene were used for crystal growth to uncover the preparation condition-mobility relationships. The maximum solubility of N@C₆₀ in *m*-xylene is around 2 mg mL⁻¹, five concentrations 0.1, 0.5, 1, 1.5 and 2 mg mL⁻¹ were used for crystal growth,

respectively. As the concentration of N@C₆₀ increased from 0.5 to 2.0 mg mL⁻¹, the mobility decreased (**Figure 2e**), while the channel coverage continued to increase. This result has also been observed for C₆₀ OFETs.³⁴ The reason for it is that the morphology of the crystals grown at high concentration is thick disordered and multi-stacked, whereas low concentrations give well-aligned crystals. In addition, crystals grown at low concentration show lower thickness. This gives a low contact resistance, which is beneficial for charge injection.

To study the charge transport mechanism of N@C60 single crystals, OFETs based on as-prown crystals were tested in vacuum when cooled from 300 to 80 K. As shown in Figure 3a-c, the μ of the N@C₆₀ single-crystal OFETs presents two different behaviors in different temperature regions. The transition temperature (T^*) between the two different regions is 180 K, and a highest μ of 2.91 cm² V⁻¹ s⁻¹, is also obtained at this temperature. Figure 3a shows the transfer curves of OFETs at the temperature from 80 K to 180 K. In the low-temperature region (80 K – 180 K), the μ increased while heating the devices, producing a positive mobility-temperature coefficient (Figure 3c). The mobility increases from 2.04 cm² V⁻¹ s⁻¹ at 80 K to 2.91 cm² V⁻¹ s⁻¹ at 180 K. Such behavior is commonly observed in organic field effect transistors,^{21,43–45} reflecting the gate voltage filling up low-mobility trap states.⁴⁶ This indicates that the charge transport through N@C60 single crystals follows a thermally activated mechanism at these temperatures.⁴⁶ In this temperature region, most of the charge carriers are trapped in localized shallow traps which are formed by chemical impurities, structural disorders, and surface states.^{42–44} Charge transport occurs *via* the extended states (transport level)

when the carriers are thermally activated and released from the traps.⁴²⁻⁴⁴ Fitting the data with the equation (1),

$$\mu = \mu_0 \exp\left(\frac{-E_a}{k_B T}\right) \tag{1}$$

where E_a is the activation energy and k_B is the Boltzmann constant, we obtained an activation energy of 6.34 meV, as shown in **Figure 3d**. The activation energy of N@C₆₀ single crystal here (6.34 meV) is comparable to the reported values for highperformance organic semiconductors, such as TIPS-PEN ($E_a = 5.7 \text{ meV}$)⁴⁴ and TIPS-TAP ($E_a = 5.6 \text{ meV}$).⁴⁷ This implies that a low degree of disorders presented in N@C₆₀ single crystals and at the interface between the single crystals and the substrates.^{46–48}

In the high-temperature region (180K – 300K), μ exhibits a negative mobilitytemperature coefficient. The field effect mobilities of N@C₆₀ OFETs decrease monotonically from 2.91 to 2.23 cm² V⁻¹ s⁻¹ by heating from 180 K to 300 K. The negative mobility-temperature coefficient is generally observed for band-like charge transport mechanism, in which charge carriers delocalize over a few molecules.^{44,45} With temperature increasing, thermal energy becomes sufficient and the influence of trapping could be eliminated. In this temperature region, the overall charge transport was dominated by the intrinsic charge transport through the delocalized state within the N@C₆₀ single crystals.^{44,45} Band-like temperature dependence is the signature of high-quality crystals and excellent electrical conduction.⁴⁸

To study the applications of $N@C_{60}$ single crystals in photodetectors, we carried out photodetection measurements on $N@C_{60}$ OFETs. In our photoconduction measurement, NIR light irradiation was provided by NIR laser diodes (780 nm). As depicted in **Figure 4a** and **4b**, with NIR irradiation on and off, the photocurrent of a N@C₆₀ OFET shows two different states. Without NIR illumination, the current was ~ 2 nA at an applied bias of 30 V. However, under a 9.5 mW cm⁻² NIR illumination, the photocurrent was as 164 nA, leading to an on/off ratio of 82. According to the time-resolved photo-response measurements (**Figure 4c**), the switch between on and off states is very sentivite with response time around 180 ms and this process is completely reversible. These excellent features enable the as-fabricated devices to act as high-performance photosensitive switches. Photoresponsivity (*R*) is a key factor to identify the light-sensitive performance of photodetectors. To further evaluate the photo-response, we calculated photo-responsivity of N@C₆₀ OFET by the equation (2):

$$R = \frac{I_{light} - I_{dark}}{PS}$$
(2)

Where I_{light} is the photocurrent when exposed to NIR light, I_{dark} is the is the dark current without illunination, P is the incident power density and S is the active area. In these devices, the active area is measured to be ~ 9.13 μ m². The specific detectivity (D^*) can be calculated by the equation (3):

$$D^* = \frac{RS^{1/2}}{(2qI_{dark})^{1/2}}$$
(3)

where q is the electron charge. We plotted the device R and D* as functions of the applied bias in **Figure 4d**. The maximum responsivity of 177.3 A·W⁻¹ was obtained at $V_{\text{GS}} = 30$ V and P = 9.5 mW cm⁻². The maximum D* of 7.5×10^9 Jones was also obtained at the same condition. However, both R and D* decreases by increasing V_{GS} beyond 30 V due to the reduced photocurrent. Both R and D* exhibit gate-modulated

property; The applied bias can effectively control the on and off states of the devices. This feature provides N@C₆₀ based photo-detectors with great potential in video frame rate imaging applications.⁴⁹ These promising results present an excellent performance of our N@C₆₀ single-crystal photodetectors in the near-infrared region, which are more satisfactory values compared with those of other reported photodetectors working under similar conditions (**Table 1**).^{13,14,50–54} The high photoresponse performance demonstrates that N@C₆₀ single crystals can serve as excellent NIR photodetection and photosensing materials.

Conclusions

In conclusion, the first demonstration of endohedral fullerene single crystals applied in OFETs and highly sensitive NIR photodetectors was reported. The maximum electron mobility of OFETs based on self-aligned N@C₆₀ single crystals is 2.23 cm² V⁻¹ s⁻¹. Photodetectors based on as-grown single crystals present highly sensitive photoconductivity properties with a responsivity of 177.3 A W⁻¹ in the near-infrared (NIR) region, which is a more satisfactory value compared with those of other reported OFETs working under similar conditions. The high mobility and highly sensitive photo-responsivity of the N@C₆₀ single crystals pave a new way for the significant advancements in OFETs and photodetection applications.

Experimental Section

Materials: C₆₀ with purity of 99.95% was purchased from SES research, n-Octadecyltrimethoxysilane (OTS) was purchased from Sigma Aldrich. *m*-xylene, toluene and other solvents were purchased from Sigma Aldrich. All materials were used without further purification. N@ C_{60} was synthesized and purified following our previous report.³³

Wafer modification: Highly doped silicon wafer with 300 nm SiO₂ layer were modified by OTS monolayer following previous report.³⁷

Crystallization: Crystals were grown *in situ* on substrates following the method in the literature.²² The concentration of N@C₆₀ in *m*-xylene is from 0.1–2.0 mg mL⁻¹.

Morphology Characterization: OM images were recorded using an Olympus BX 60 optical microscope. AFM images were recorded in a NT-MDT Ntegra atomic force microscope in semi-contact mode. SEM images were recorded by an FEI Inspect-F scanning electron microscope. XRD pattern was obtained by a Rigaku D/max-2500 X-ray diffractometer using filtered Cu K α radiation ($\lambda = 1.54$ Å) and the 2 θ is ranging from 5° to 40°. The XRD pattern was firstly compared to the reported XRD pattern of C₆₀ nanorod (a = b = 23.76 Å, c = 10.08 Å) in the literature.⁴¹ We then calculated the cell dimensons according to Bragg's Law. TEM images were recorded in a JEOL JEM-2010 transmission electron microscope in a high-resolution mode and electron diffraction configuration, at an accelerating voltage of 200 kV. The single crystals were grown on the carbon coated mica substrates by the method mentioned above. Then carbon film were put onto the water surface and then copper grids were employed to pick up the single crystals. The samples were annealled under vacuum for 24 hours and then go for TEM observation.

Device fabrication and measurement: OFETs were fabricated in a top-contact bottomgate device configuration by depositing 50 nm Au as source and drain electrodes. Current–voltage characteristics of the devices were carried out under a vacuum of 10^{-6} Torr on a Lake Shore model PS-100 tabletop cryogenic probe station with a Keithley 4200-SCS semiconductor characterization system. The field-effect mobility was calculated in the saturation regime according to the equation $I_{DS} = (\mu W C_i/2L)(V_G - V_T)^2$, where I_{DS} is the drain–source current, μ is the field-effect mobility, W is the channel width, L is the channel length, C_i is the capacitance per unit area of the gate dielectric layer, V_G is the gate voltage and V_T is the threshold voltage. The channel length (L) is 25 mm and the channel width (W) was obtained from the SEM images.

Temperature-dependent measurement: The current-voltage measurement was carried out on the same Lake Shore model PS-100 tabletop cryogenic probe tation at a background pressure of 1×10^{-6} torr. The cooling process from 300 K to 80 K was realized by introducing liquid nitrogen into the insulation cabin outside the sample cabin. Temperature of the samples was balanced by simultaneous heat under the sample holder.

Photoresponse Measurement: Photoresponse measurements were performed on the same Lake Shore model PS-100 tabletop cryogenic probe tation by two-terminal mode. The devices were put in a sample chamber under a vacuum of 10^{-6} Torr at room temperature. The parameters were analysed using a Keithley 4200-SCS semiconductor characterization system.

ASSOCIATED CONTENT

Supporting Information.

XRD pattern of as-prepared N@C₆₀ single crystals. HRTEM image of as-prepared N@C₆₀ single crystal. Typical gate-bias dependent mobility for devices based on N@C₆₀ needle crystals. Channel width dependences of mobilities in OFETs based on N@C₆₀ single crystals.

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Author Contributions

X.Z. and T.L. conceived the idea and design of the experiments. X.Z. and T.J.S.D performed the synthesis and purification of N@C₆₀. X.Z. and T.L. performed the single crystal growth of N@C₆₀. X.Z. and X.H. carried out the image collection and result analysis of OM. X.Z. and W.S. carried out the image collection and result analysis of SEM. X.Z. and Z.L. carried out the image collection and result analysis of AFM. X.Z. and T.L. performed the measurements and result analysis of TEM and SAED. X.Z. and T.L. fabricated the OFET devices and performed the temperature-dependent measurements and data analysis. X.Z and W.S. carried out the photo-response studies. X.Z. wrote the manuscript and all authors contributed to the revision of the manuscript. T.J.S.D. supervised this work.

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Notes

The authors declare no competing financial interest.

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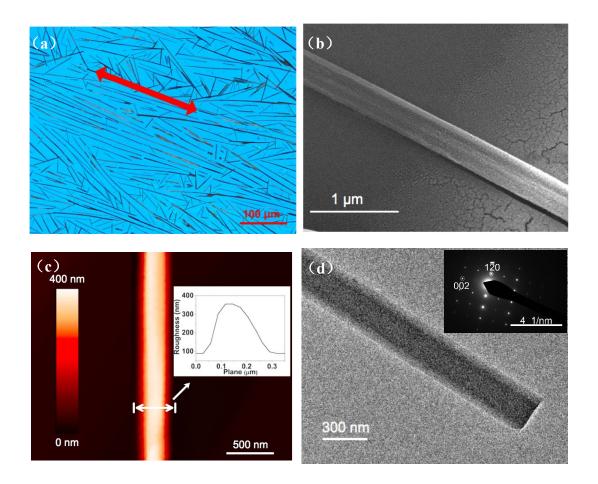


Figure 1 (a) OM and (b) SEM images of self-aligned N@C₆₀ needle crystals; (c) AFM image of a needle crystal showing facted shape; (inset) selected area roughness analysis; (d) TEM image of a needle crystal; (inset) SAED pattern containing a single set of spots, indicating the crystallinity of needle.

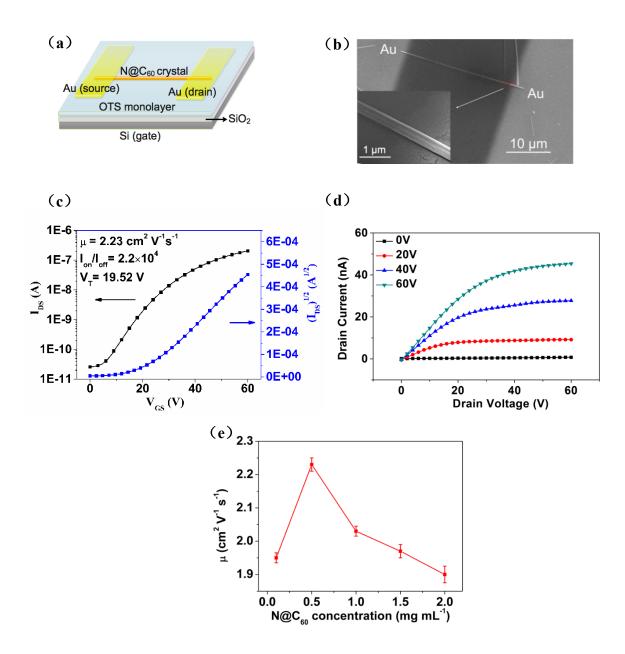


Figure 2 (a) Schematic of a typical device; (b) SEM image showing N@C₆₀ needle crystal between source (S) and drain (D) electrodes, channel length (L) was measured from the real channel length and channel width (W) was measured from the contacting area of the crystals that cross the S and D electrodes; (c) and (d) Typical transfer and output characteristics of the FETs based on N@C₆₀ needle crystals grown from a solution with a concentration of 0.5 mg mL⁻¹; (e) OFET mobilities as a function of the N@C₆₀ concentrations for crystallization.

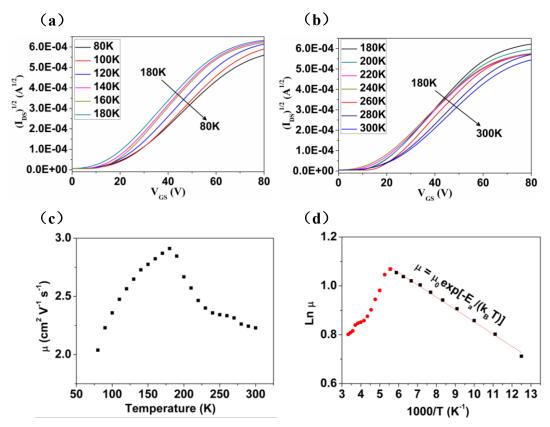


Figure 3 Square root of drain current $(I_{DS})^{1/2}$ versus gate voltage (V_{GS}) for the N@C₆₀ needle crystal OFETs with a drain voltage of 80 V as measured at the temperature from (a) 80 to 180K and (b) 180 to 300K; (c) temperature dependence of field effect mobility (μ) for N@C₆₀ single crystal OFETs; (d) ln(μ) versus 1/T for N@C₆₀ single crystal OFETs.

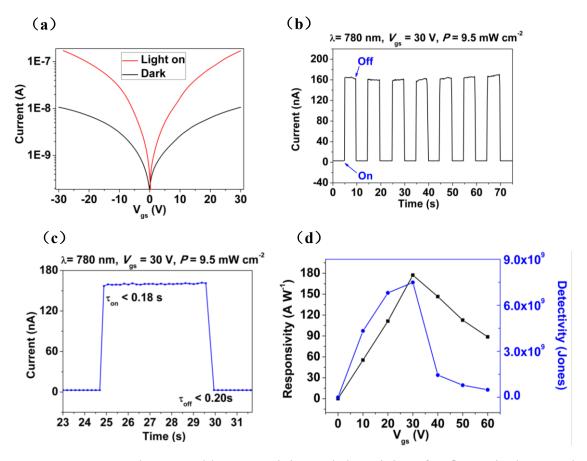


Figure 4 Gate-voltage-tunable responsivity and detectivity of N@C₆₀ single crystal photodetectors. (a) I-V characteristics of the N@C₆₀ needle crystals with and without NIR irradiation at a light density of 9.5 mW cm⁻²; (b) Time-resolved photoresponse was measured at 780 nm, which indicates a highly stable and reversible response of the phototransistor; (c) Temporal response exhibits fast photoswitching ($\tau_{on} < 0.18 \text{ s}, \tau_{off} < 0.20 \text{ s}$); (d) Gate-tunable behavior for responsivity and detectivity with *P* = 9.5 mW cm⁻².

Material	Wavelength	Power density	Responsivity	Reference
	(nm)	(mW cm ⁻²)	(A W ⁻¹)	
BPTT	380	1.55	82	50
pentacene	365	5.0	50	51
	650	5.0	0.45	
Spiro-4p-CPDT	370		25	52
C8BTBT	473	2.7	33	13
C_{60}	780	1.5	82.6	24
C ₆₀	470	0.000523	1047	56
C ₆₀	360	4.38	75.3	14
	650	4.38	90.4	
F8T2	405	3	0.0004	53
graphene	532		8.61	54
N@C ₆₀	780	9.5	177.3	In this work

 Table 1 Comparison of performance for photodetectors based on various organic

 semiconductors

Table of Contents Graphic

